Optical absorbance in multilayer two-dimensional materials: Graphene and antimonene

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ABSTRACT

Antimonene, one of the group V elemental monolayers, has attracted intense interest due to its intriguing electronic properties. Here, we present the optical absorption properties of atomically flat antimonene for which the directional bonds between Sb atoms appear to be analogous to C–C bonds in graphene. The results, based on first-principles density functional theory calculations, predict the absorbance in multilayer antimonene to be comparable or higher than that calculated for multilayer graphene. Specifically, the IR absorption in antimonene is significantly higher with a prominent band at about 4 μm associated with the dipole-allowed interband transitions. Furthermore, a strong dependence of absorbance on topology is predicted for both antimonene and graphene which results from the subtle variations in their stacking-dependent band structures. Our results suggest multilayer antimonene to be a good candidate material for optical power limiting applications in the IR region.

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The optical properties of two-dimensional (2D) atomically thin layers have attracted a great deal of attention in recent years. From the early measurements of nonlinear transmission and scattering and two-photon absorption (TPA) on graphene and four-wave mixing measurements on graphene flakes to the measurements of the intensity-dependent refractive index of antimonene and more recently 2D GeP, there has been a surge in the experimental observations and theoretical calculations on the linear and nonlinear optical (NLO) properties of 2D layered materials. Several excellent reviews on the subject have recently appeared in the literature. Among the 2D materials, such as graphene, boron nitride (BN), phosphorene, and antimonene, the latter offers several appealing properties for applications in optoelectronics and photonics devices. The theoretical calculations have predicted 2D antimonene to be a narrow bandgap semiconductor with two configurations of monolayer antimonene, α-Sb, which has a puckered structure with two atomic sublayers with an indirect bandgap of ~0.28 eV and β-Sb with a buckled hexagonal lattice and a slightly higher indirect bandgap of ~0.76 eV. Note that β-Sb can be synthesized by various physical or chemical methods such as molecular-beam epitaxy, epitaxy growth, etc. Moreover, unlike phosphorene, atomically flat antimonene can be grown on a lattice-matched Ag (111) substrate with hexagonal symmetry. It, therefore, appears that sp2 hybridized Sb atoms form a planar graphene-like antimonene in which the lone-pair electrons on adjacent sites likely form out-of-plane nonbonding orbitals (Fig. S1).

Since directional bonding between Sb atoms in the planar antimonene appears to be analogous to C–C bonds in graphene, it is of interest to examine the optical properties of multilayer antimonene vis-à-vis multilayer graphene. Here, we present the optical absorption spectra ranging from near-infrared (0.8–5.0 μm) to visible (400–740 nm). Both layer- and topology-dependence on the absorption spectra are investigated via a detailed analysis of the band structure.

Calculations were performed by the density functional theory (DFT) approach using the projector augmented wave (PAW) potentials as implemented in the Vienna ab initio simulation package (VASP). The exchange and correlation functionals were treated within the framework of generalized gradient approximation (GGA) given by the Perdew–Burke–Ernzerhof (PBE) functional. Contributions from the van der Waals interactions were incorporated by using the Grimme’s semi-empirical dispersive D2 term. Additional details are given in the supplementary material, e.g., Figs. S2–S4.
Low-energy optical absorption can be measured via absorption spectroscopy\(^{18-24}\) and calculated using the imaginary part of the dielectric function\(^{25-28}\). Here, the frequency-dependent dielectric matrix ($\varepsilon_2(\omega)$) was calculated by the “sum-over-states” (SOS) method.\(^3\) The fraction of photon energy ($E = h\omega$) absorbed by a multilayer is given as

$$ A(\omega) = \frac{\omega}{c} \varepsilon_2(\omega) L_z, \quad (1) $$

where $L_z$ is the length of a supercell in the $z$-direction and $c$ is the speed of light. In this way, the calculated absorbance is independent of the size of a vacuum in the $z$-direction of the periodic supercell (Fig. S5, supplementary material), and is equivalent to the absorbance calculated using the expression \(\frac{\varepsilon_2(\omega)}{\varepsilon_1(\omega)}\)\(^3\) with $\varepsilon_2$ being the imaginary part of the polarizability per unit area. For a multilayer system, thickness is equivalent to $L_z$, and is defined as a sum of the vacuum distance and interlayer distance. The dependence of $\varepsilon_2(\omega)$ on the vacuum distance considered in calculations is displayed in Fig. S4, supplementary material. Note that dielectric functions are inversely proportional to the volume of the cell in a periodic super-cell method [Eq. (1), supplementary material].

DFT calculations were first performed on multilayer graphene for which the experimental low-energy absorption spectrum is known in the literature.\(^{11,30}\) In our work, we find the energetically preferred configurations to be AB-stacked for bilayer (2L), ABA-stacked for trilayer (3L), and ABAB-stacked for a quad-layer (4L) at the DFT (PBE) + D2 level of theory (Table S1, supplementary material). It is worth noting that multilayer graphene exhibits distinct optical conductivity for Bernal (ABA) and rhombohedral (ABC) stacking in infrared absorption spectroscopy measurements.\(^2\) Further, trilayer-graphene also exhibits distinct electronic structures for different stackings (AAA, ABA, and ABC) as measured with NanoARPES\(^3\) and calculated using the tight-binding method.\(^3\)

The calculated optical absorbance properties of multilayer graphene in terms of absorbance are shown in Fig. 1. Here, 1L is a monolayer, 2L is an AB-stacked bilayer, 3L is an ABA-stacked trilayer, and 4L is ABAB-stacked quadlayer graphene. In the IR region, the absorbance of monolayer graphene is nearly constant (~2.3%), which agrees well with experiments\(^1\) and the calculated results obtained using the imaginary part of polarizability per unit area.\(^2\)

For graphene, the out-of-plane polarization component of the electromagnetic field for absorbance is nearly zero in the visible spectrum, suggesting that it is optically transparent (Fig. S5, supplementary material). On the other hand, the in-plane polarization component of the electromagnetic field for absorbance is large and is mainly attributable to the atomic thickness of monolayer graphene. The most prominent absorption band appears at ~4 eV (~310 nm) (Fig. S5, supplementary material), which is consistent with the results of previous studies.\(^2\) The increase in the intensity of the ~4 eV band with the number of layers is likely due to the availability of a larger number of bands associated with the interband transition at the M point (Fig. S6, supplementary material).

In the 1.77–3.1 eV (i.e., 700–400 nm) region, a relatively small increase in transmittance is predicted [Fig. S7(b), supplementary material] and is consistent with experiments.\(^18,31,33\) As thickness increases, the nonlinear behavior of transmittance can be seen in multilayer graphene. For $\lambda = 300$ nm and 800 nm [Figs. S7(c) and S7(d), supplementary material], a linear decrease in transmittance in going from monolayer to bilayer is noticed. Beyond $n=2$, the transmittance shows a small nonlinearity which is relatively pronounced for 300 nm. The calculated results affirm the observation of the nonlinear behavior of transmittance of white light for five-layer graphene.\(^7\) Overall, the transmission was reduced to ~79% for 800 nm, ~74% for 532, and ~40% for 300 nm in ABAB-stacked 4L relative to that of the monolayer.

Next, we calculate the frequency-dependent refractive index given by the expression, \(n(\omega) = \frac{1}{\sqrt{\varepsilon_1(\omega)}} \left[ \frac{1}{2} \left( \frac{\varepsilon_2(\omega)}{\varepsilon_1(\omega)} \right)^{-1} \right]^{1/2} \). The real part of dielectric function ($\varepsilon_1$) is obtained from the imaginary part of dielectric function ($\varepsilon_2$) using the Kramers–Kronig relations.\(^3\) \(n(\omega)\) exhibits a large increase in going from single to bilayer, then shows reduced increases upon successive addition of layers (Fig. S8, supplementary material).

The planar antimonene is semi-metallic\(^3\) and, unlike graphene, its absorption spectrum shows a relatively larger increase in absorbance in going from the near-IR to visible region (Fig. 2). Absorption increases linearly with photon energy, though a small absorption band appears at 1.2 eV (≈1 μm) with an absorbance of nearly 4%. This band is attributed to the interband electronic transition from the valence band maximum (VBM) to the conduction band minimum (CBM) along the M–K–Γ direction. Note that the buckled antimonene shows negligible absorption in the visible region, whereas relatively small visible light absorption is predicted for multilayer buckled antimonene.\(^3\) For (atomically flat) multilayer antimonene, Table S2 (supplementary material) shows the stacking configurations to be degenerate with $\Delta E \lesssim 13$ meV, although the AA-stacked bilayer and the AAA-stacked trilayer can be considered as the energetically preferred configurations. Previously, the buckled multilayer antimonene was predicted to prefer ABC-stacking, though decreasing its interlayer separation, leads to either AAA- or ABAB-stacking configurations with the energy difference of 113 meV/atom.\(^3\)

We find that the calculated transmittance decreases as the thickness increases in going from monolayer to trilayer (Fig. S9, supplementary material). The decrease in transmission in the trilayer relative
to the monolayer is about $\approx 25\%$ for 400 nm and about $\approx 10\%$ for 800 nm. The calculated refractive index of the planar antimonene varies with the thickness (Fig. S10, supplementary material) in contrast to the results obtained for multilayer graphene. At 400 nm, the calculated refractive index is 2.12 for monolayer and 2.2 for bilayer/trilayer antimonene. Note that the refractive index of the buckled $\beta$-antimonene is reported to be 2.0 for the UV spectral region. 49

In multilayer graphene, a weak interlayer interaction allows a variety of stacking patterns to be formed; bilayer graphene can be arranged in either AB (Bernal) or AA (hexagonal) stacking configurations, whereas trilayer graphene may exist in ABA (Bernal), ABC (rhombohedral), or AAA (hexagonal) stacking configurations. 49 ABA-stacked trilayer graphene is predicted to be energetically stable as compared to ABC and AAA stacked trilayers. The energy difference ($\Delta E$) is $\approx 4$ meV and $\approx 51$ meV for ABC- and AAA-stacked trilayers, respectively. Despite a rather small $\Delta E$ among the three stacking configurations, noticeable differences are seen in their IR absorbance, which suggests that the optical response can be influenced by topology (Fig. 3). For example, the ABC-stacked trilayer shows a prominent absorption band at 0.45 eV ($\approx 2.75 \mu m$) with absorbance significantly higher than that of the AAA-stacked trilayer (Fig. 3, inset). It should be noted that thermal effects, which were not included in the present study, can be a factor in modifying absorbance below 0.1 eV ($\approx 12 \mu m$).

To obtain further insight into the stacking-dependent absorption, we also calculated the low-energy electronic band structure around the K-point (Fig. S11, supplementary material). Here, the AAA-stacked trilayer shows a distinctly different electronic structure than the ABA or ABC stacked trilayer. The band structure of AAA-stacked graphene can be seen as a superposition of bands associated with individual layers, showing three Dirac cones near the Fermi level (Fig. S11(c)). Due to a highly symmetric stacking pattern, the middle layer of graphene interacts equally with the top and the bottom layer, which results in a delocalized charge distribution associated with the middle layer. These results are in agreement with those reported in previous theoretical studies. 20,22,23,41,42 There exist two pairs of energy bands associated with VBM and CBM near the Fermi level for ABA stacking, and only one pair for ABC stacking. The other pairs of bands associated with VBM and CBM move away from the Fermi level which may lead to the prominent absorption at 0.45 eV as seen in Fig. S11(b). These stacking-dependent features may likely be associated with the number of uncoordinated atomic sites in adjacent layers; ABA stacking has two equidistant layers (top and bottom layers). This is reflected in the calculated charge density difference, $\Delta \rho = \rho_T - (\rho_{AB/AA} + \rho_{A/C})$ where $\rho_T$ is the total charge density of the trilayer system, $\rho_{AB/AA}$ is the charge density of the AB or AA bilayer, and $\rho_{A/C}$ is the charge density of the top A or C layer of trilayer graphene (Fig. S12, supplementary material). A weak interlayer coupling between layers is seen in the charge density difference profile.

The topological dependence of the optical absorption calculated for trilayer antimonene is shown in Fig. S13. The absorption band in the visible region is at about 2.2 eV ($\approx 0.56 \mu m$) in AAA, 2.3 eV ($\approx 0.54 \mu m$) in ABA, and 2.0 eV ($\approx 0.62 \mu m$) in ABC stacked configurations, while a noticeable IR absorption at $\approx 0.5$ eV ($\approx 2.5 \mu m$) appears only for ABA-stacked antimonene. To understand these features, we examine the low-energy electronic band structure along the $\Gamma$–K–M direction together with the partial density of states (Figs. S13–S15, supplementary material). The $\approx 2.5 \mu m$ bands are likely associated with the interband transition along the K–M direction for ABA-stacked antimonene. Analysis of the partial density of states suggests this transition be associated with the electric dipole allowed $s \rightarrow p$ transition in ABA-stacked antimonene. Note that a significant increase in the number of bands available for transition in trilayer antimonene leads to higher absorption compared to that calculated for the bilayer antimonene (Fig. S16, supplementary material).

The low-energy absorption calculated for graphene and antimonene is compared in Fig. 4. Although, graphene and antimonene both are semi-metallic, a significant difference in their optical absorption is expected since the electronic configurations of C: [He] 2s22p2 and Sb: [Kr] 4d105s25p3 differ significantly. In graphene, each C atom forms a covalent ($\sigma$) bond with each of its neighbors together with a $\pi$-bond oriented out-of-plane. On the other hand, in the case of antimonene,
the out-of-plane component is the lone-pair electron on Sb atoms (Fig. S1, supplementary material), which may interact weakly with neighboring atoms as compared to the π-orbital interaction in the case of graphene, leading to the calculated difference in the absorption spectra of graphene and antimonene.

The difference in the electronic configurations of C and Sb is reflected in absorption in the visible spectral region where antimonene absorbs significantly higher than graphene; at 400 nm, absorbance is 3% and 9% for graphene and antimonene, respectively. The difference is further enhanced in the IR spectral region where a prominent, broadband appears at ≈4 μm for trilayer antimonene. This is not the case with trilayer graphene, for which a nearly linear decrease in absorption is predicted ranging from 2 μm to 15 μm. This suggests that multilayer antimonene could be a good candidate as a mid-IR tunable laser absorber.

In summary, absorbance in (atomically flat) multilayer antimonene is calculated to be comparable or greater than that for multilayer graphene. Specifically, the IR absorption in antimonene is significantly higher than that in graphene, with a prominent band at about 4 μm associated with the dipole-allowed interband transitions in the trilayer (atomically flat) antimonene. Overall, our study suggests multilayer graphene and antimonene are good optical absorbers, especially in the infrared region of the spectrum, and can potentially be used as a coating for tunable laser absorber.

See the supplementary material for the molecular orbitals in graphene and antimonene; computational details—the convergence of k-point grid and vacuum distance; and results (figures and tables)—multilayer graphene: absorbance and transmittance, band structure, and refractive index; multilayer antimonene: absorbance and transmittance, refractive index, band structure, density of states, and charge density.

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**DATA AVAILABILITY**

The data that supports the findings of this study are available within the article (and its supplementary material).

**REFERENCES**